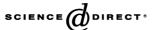


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Fatigue crack growth analysis of pre-strained 7475–T7351 aluminum alloy

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Abstract

Aluminum alloys are widely used in aeronautical design due to their good mechanical properties and low densities. Among these alloys, 7475 (Al–Zn–Mg–Cu), modified from 7075, is successfully used due to strength similar to that of 7075 and superior fracture toughness. In this study, strips of 7475–T7351 were subjected to two tensile pre-strain levels of 3 and 5%. Using compact tension C(T) specimens, fatigue crack growth (FCG) tests were conducted in air under constant amplitude loading at a stress ratio of 0.5. Three FCG rate models, namely Collipriest, Priddle, and modified Forman were examined. The results showed that both fracture toughness and FCG resistance decreased with an increase in the prestrain level. Both Collipriest and Priddle models fit the FCG rate data in a similar fashion. However, the Priddle model provided a better fit than Collipriest. The modified Forman model provided the best fit to the FCG rate data as compared with the other two models. The modified Forman model used to fit other unpublished FCG data and its response was also superior to the others. Therefore, this model may be suggested for use in critical applications, such as aeronautical structural design.

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Keywords: 7475-T7351; Pre-strain; Fatigue crack growth rate; Statistical analysis; Modeling

1. Introduction

High strength age-hardenable aluminum alloys, such as 2xxx and 7xxx, have been the most widely used structural materials in aeronautical industries due to their good mechanical properties and low densities [1–8].

The copper and magnesium bearing alloys (2xxx) are widely used for aircraft and space vehicles. The naturally aged and cold worked 2024-T3 is the most popular alloy of this group due to its good damage tolerance [2–5]. However, in the form of thick sheet, this alloy is susceptible to exfoliation corrosion [2,9].

The zinc and magnesium bearing alloys (7xxx), of which 7075 is the most widely used, have the highest strengths by far. In the conventional peak-aged temper (T6), the thick plates, forgings, and extrusions of 7xxx series alloys are susceptible to stress corrosion cracking (SCC) [2,9–11]. The over-aged

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temper (T7) has been developed to improve the SCC resistance; however, this results in loss of strength compared with the T6 temper [12,13].

The need for high strength coupled with good fracture toughness and SCC resistance led to the development of 7475 alloy [14]. This progress resulted from a considerable reduction in the levels of iron, silicon, and manganese, but also from improvements in thermomechanical and heat treatment practices [2,15].

In some cases, 7475–T7351 is being used after application of certain levels of pre-strain. Regarding the influence of pre-straining on FCG behavior, few works [16–19] have been published. Hence, the objectives of this study are to: (a) show the effect of pre-strain on FCG rate in 7475–T7351, and (b) test different FCG models to determine which model provides the best-fit to the experimental data. To achieve these objectives, FCG data of 7475–T7351 subjected to two tensile pre-strain levels of 3 and 5% were collected. Due to the sigmoidal shape of the $\log(\frac{da}{dN})-\log(\Delta K)$ curve, three models, namely, Collipriest [20,21], Priddle [22], and modified Forman [23] were chosen. They model all three regions of the FCG rate curve. Nonlinear statistical analyses [24,25] were done to estimate the model parameters and to show which model best fits the FCG rate data.

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Table 1 Chemical composition of 7475 aluminum alloy (wt%)

Alloy	Si	Fe	Cu	Mn	Mg	Zn	Cr	Ti	Al
7475	0.029	0.085	1.661	0.01	2.376	5.722	0.21	0.0266	Balance

2. Material and experimental procedure

2.1. Material and specimens

Aluminum alloy plate of 7475–T7351 with a thickness of 76.2 mm was used in this study. The chemical composition is shown in Table 1. Strips of 6 mm thickness, shown in Fig. 1, were machined from the original plate and then subjected to two tensile pre-strain levels of 3 and 5%. From these strips, tensile test specimens, illustrated in Fig. 2, were machined in the longitudinal direction, according to ASTM E8 [26]. The C(T) specimens used for fracture toughness and FCG tests were machined in L-T direction, according to ASTM E647 [27]; the specimen dimensions are shown in Fig. 3.

2.2. Fatigue crack growth testing

FCG tests were conducted on pre-cracked C(T) specimens using the method described in ASTM E 647 [27]. FCG tests were done under a constant amplitude sinusoidal wave loading at a stress ratio (R) of 0.5. This stress ratio was suggested to reduce the crack closure effect [28]. FCG tests were conducted using a 100 kN MTS servo-hydraulic testing machine interfaced to a computer for machine control and data acquisition. All tests were conducted at a frequency of 30 Hz under room temperature conditions that ranged from 20 to 25 °C with a relative humidity from 60 to 70% in air. The crack length was measured using a compliance method, in which a clip gauge is used to measure the elastic compliance of the specimen, which tends to increase with crack growth.

In addition, FCG threshold ($\Delta K_{\rm th}$) tests were done using a 'load shedding' method proposed in ASTM E647 [27]. Basically, these tests were conducted where the stress intensity factor range ($\Delta K = K_{\rm max} - K_{\rm min}$) was decreased. Consequently, crack growth slowed down and the FCG threshold was reached as the crack stopped growing, or reached a sufficiently low

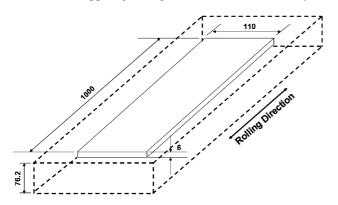


Fig. 1. Dimensional details of 7475–T7351 aluminum alloy plate and the strip (dimensions in millimeter).

FCG rate. According to ASTM E647, ΔK_{th} is about the ΔK corresponding to a FCG rate (da/dN) of 10^{-10} m/cycle.

2.3. Fracture toughness testing

Fracture toughness tests were conducted on the C(T) specimens, using an MTS servo-hydraulic testing machine, according to ASTM E561 [29]. Basically, cyclic loading was applied to introduce a fatigue crack. As the crack reached the desired length, the fatigue cycling was stopped, and the load was gradually increased until fracture occurred. The stress intensity at fracture was calculated as the fracture toughness. All tests were done at room temperature and in air environment.

3. Results and discussions

3.1. Tensile test

The results of tensile tests are presented in Table 2. Both 0.2% yield and ultimate tensile strength increase with an increase in the pre-strain level from 0 to 5%. On the other hand, the total strain to fracture decreases. These results are expected due to the effect of strain-hardening [30,31].

3.2. Fatigue threshold and fracture toughness

Table 3 shows the results of the threshold stress intensity factor range ($\Delta K_{\rm th}$) and fracture toughness ($K_{\rm c}$) tests of 7475–T7351 for all cases of pre-strain studied. The fracture toughness decreases with an increase in the pre-strain level. This is expected and is attributed to the effect of strain-hardening that leads to an increase in strength of the material. When the strength increases, the strain-hardening exponent decreases. This favors the formation and nucleation of microvoids with void sheet occurrence. This limits crack-tip opening displacement, thereby decreasing fracture toughness [32]. However, a general relation between the fatigue threshold and pre-strain was not found, since fatigue threshold is sensitive to many intrinsic and extrinsic features [33,34].

3.3. Fatigue crack growth rate modeling

3.3.1. FCG rate models

The reason for building models is to link theoretical ideas with the observed data. Modeling of FCG rate data has enhanced the ability to create damage tolerant design philosophies. Paris and Erdogan [35] proposed the most important and popular work. They were the first who correlated FCG rate with fracture mechanics parameters (K_{\min} and K_{\max}), describing the loading conditions in the region of the crack

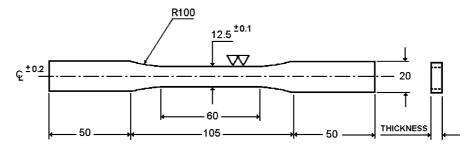


Fig. 2. Dimensional details of the tensile test specimen (dimensions in millimeter).

front. They observed a linear relationship between FCG rate (da/dN) and ΔK when plotted on a log-log scale. Paris and Erdogan proposed the power law relationship:

$$\frac{\mathrm{d}a}{\mathrm{d}N} = C\Delta K^n \tag{1}$$

where C and n are material parameters that are determined experimentally.

The Paris–Erdogan equation does not consider: (a) the effect of stress ratio, (b) the existence of fatigue threshold, and (c) the accelerated FCG rate when the maximum stress intensity factor (K_{max}) approaches material fracture toughness (K_c) . It does not adequately describe region I or III FCG rates; it tends to overestimate region I and underestimate region III FCG rates. Although the Paris–Erdogan equation is a simplification of a very complex phenomenon, it is still very popular on account of significant engineering interest.

The typical curve of a $\log(da/dN)-\log(\Delta K)$, at a prescribed condition (environment and stress ratio), is sigmoidal in shape. It comprises three regions and is bounded at its extremes by the threshold stress intensity factor range and the critical stress intensity factor range. In the intermediate region of the curve,

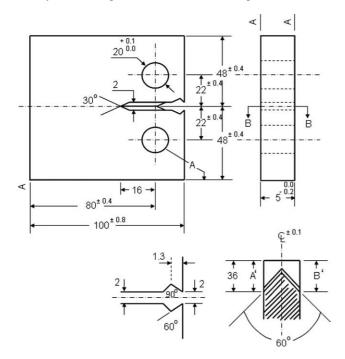


Fig. 3. Dimensional details of the C(T) specimen (dimensions in millimeter).

there is a linear relation between $\log(da/dN)$ and $\log(\Delta K)$, as proposed by Paris and Erdogan.

Based on Eq. (1), different FCG rate models have been proposed to fit all or part of the sigmoidal curve. In this study, three models that fit all parts of FCG curve are adopted. These models are:

1. Collipriest model

Collipriest proposed the following model [20,21]:

$$\log \frac{\mathrm{d}a}{\mathrm{d}N} = C_1 + C_2 \tanh^{-1} \left[\frac{\log \left(\frac{\Delta K^2}{\Delta K_{\mathrm{th}} K_c (1-R)} \right)}{\log \left(\frac{K_c (1-R)}{\Delta K_{\mathrm{th}}} \right)} \right] \tag{2}$$

where

$$C_1 = \log(C(K_c \Delta K_{th})^{n/2})$$
 and $C_2 = \log\left(\frac{K_c}{\Delta K_{th}}\right)^{n/2}$

2. Priddle model

Priddle proposed the following model [22]:

$$\frac{\mathrm{d}a}{\mathrm{d}N} = C \left(\frac{(\Delta K - \Delta K_{\mathrm{th}})(1 - R)}{((1 - R)K_{\mathrm{c}} - \Delta K)} \right)^{n} \tag{3}$$

Table 2 Average room temperature tensile test properties of 7475–T7351 for all prestrain cases

Pre-strain level	Orient.	$\sigma_{\rm YS}$ (MPa)	σ_{TS} (MPa)	ε (%)
0%	L	405.76	482.60	12.11
3%	L	412.17	496.38	9.82
5%	L	416	497.67	8.82

 $\sigma_{\rm YS},\,0.2\%$ yield tensile strength; $\sigma_{\rm TS}$, ultimate tensile strength; ε , total strain.

Table 3
Fatigue threshold and fracture toughness values of 7475–T7351 for all prestrain cases

Pre-strain level (%)	$\Delta K_{\rm th}~({\rm MPa}~{\rm m}^{0.5})$	K _c (MPa m ^{0.5})
0	1.31	95.5
3	1.45	92.8
5	1.37	79.3

Table 4
Statistical analysis results of 7474–T7351 FCG rate data for 0% pre-strain, R=0.5, comparison among Collipriest, Priddle, and modified Forman models

Model	Parameter	Estimate	Standard error	<i>t</i> -value	<i>p</i> -value	95% LCI	95% UCI
Collipriest	С	2.364E-09	5.091E-10	4.6428	< 0.001	1.355E-09	3.372E-09
	n	1.407	0.05145	13.3837	< 0.0001	1.199	1.615
	Se	0.3279	R^2	0.8895	F-value	934	
	SSF	12.4697	R_{adj}^2	0.8886	<i>p</i> -value	< 0.0001	
Priddle	C	3.587E-06	6.41E-07	5.9377	< 0.0001	2.390E-06	4.783E-06
	n	1.558	0.044507	34.9991	< 0.0001	1.470	1.646
	Se	0.2900	R^2	0.9135	F-value	1224.9	
	SSE	9.7551	$R_{\rm adj}^2$	0.9127	<i>p</i> -value	< 0.0001	
Modified For-	C	4.003E-09	2.628E-10	15.2308	< 0.0001	3.482E-09	4.52E-09
man							
	n	2.582	0.039212	65.8420	< 0.0001	2.504	2.659
	Se	0.1323	R^2	0.9819	F-value	6301.4	
	SSE	2.0342	$R_{ m adj}^2$	0.9817	<i>p</i> -value	< 0.0001	

SSE, sum of squares of error; Se, standard error of the residual; R^2 , coefficient of determination; R^2_{adj} , adjusted coefficient of determination; LCI, lower confidence interval; UCI, upper confidence interval.

Table 5
Statistical analysis results of 7474–T7351 FCG rate data for 3% pre-strain, R=0.5, comparison among Collipriest, Priddle, and modified Forman models

Model	Parameter	Estimate	Standard error	t-value	<i>p</i> -value	95% LCI	95% UCI
Collipriest	С	1.178E-09	1.466E-10	7.9958	< 0.0001	8.31E-10	1.461E-09
•	n	1.910	0.052626	36.2871	< 0.0001	1.806	2.013
	Se	0.2394	R^2	0.9462	F-value	4223.2	
	SSE	137571	$R_{ m adj}^2$	0.9460	<i>p</i> -value	< 0.0001	
Priddle	C	8.754E-06	6.278E-07	13.9433	< 0.0001	7.517E-06	9.991E-06
	n	1.750	0.023382	75.8493	< 0.0001	1.704	1.796
	Se	0.2068	R^2	09589	F-value	5602.5	
	SSE	102672	$R_{ m adj}^2$	0.9587	<i>p</i> -value	< 0.0001	
Modified For-	C	3.560E-09	1953E-10	18.2214	< 0.0001	3.17E-09	3.94E-09
man							
	n	2.720	0.026429	102.9073	< 0.0001	2.668	2.772
	Se	0.1265	R^2	0.9846	F-value	15339.7	
	SSE	3.8460	$R_{ m adj}^2$	0.9845	<i>p</i> -value	< 0.0001	

SSE, sum of squares of error; Se, standard error of the residual; R^2 , coefficient of determination; R^2_{adj} , adjusted coefficient of determination; LCI, lower confidence interval; UCI, upper confidence interval.

Table 6
Statistical analysis results of 7474–T7351 FCG rate data for 5% pre-strain, *R*=0.5, comparison among Collipriest, Priddle, and modified Forman models

Model	Parameter	Estimate	Standard error	t-value	<i>p</i> -value	95% LCI	95% UCI
Collipriest	С	1.726E-09	3.944E-10	1.3770	< 0.001	9.427E-10	2.510E-09
_	n	1.772	0.097834	18.1142	< 0.0001	1.578	1.967
	Se	0.2655	R^2	0.9367	F-value	1317.5	
	SSE	6.2730	$R_{ m adj}^2$	0.9360	<i>p</i> -value	< 0.0001	
Priddle	C	5647E-06	6.664E-07	8.4730	< 0.0001	4.332E-06	6.971E-06
	n	1.665	0.039927	41.6928	< 0.0001	1.585	1.744
	Se	0.2313	R^2	0.9513	F-value	1738.3	
	SSE	4.7606	$R_{ m adj}^2$	0.9507	<i>p</i> -value	< 0.0001	
Modified For-	C	3.946E-09	3.588E-10	10.9993	< 0.0001	3.23E-09	4.66E-09
man							
	n	2.667	0.044118	60.4521	< 0.0001	2.579	2.755
	Se	0.1329	R^2	0.9838	F-value	5412.5	
	SSE	1.5711	$R_{ m adj}^2$	0.9836	<i>p</i> -value	< 0.0001	

SSE, sum of squares of error; Se, standard error of the residual; R^2 , coefficient of determination; R^2_{adj} , adjusted coefficient of determination; LCI, lower confidence interval; UCI, upper confidence interval.

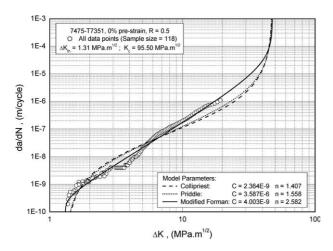


Fig. 4. FCG rate curves of 7475–T7351, 0% pre-strain, R = 0.5, comparison among Collipriest, Priddle, and modified Forman models.

3. Modified Forman model

Forman proposed its modified model [23] as:

$$\frac{\mathrm{d}a}{\mathrm{d}N} = \frac{C(1-R)^m \Delta K^n (\Delta K - \Delta K_{\mathrm{th}})^p}{((1-R)K_c - \Delta K)^q} \tag{4}$$

Certain values of the exponents m, p, and q of Eq. (4) give other forms of the crack growth equation as indicated below:

Exponent value	Equation form
m=p=q=0	Paris
m=p=0 and $q=1$	Forman
$p = q = 0$ and $m = (m_{\rm w} - 1)^{\rm n}$	Walker

Collipriest, Priddle, and modified Forman models are nonlinear regression models. A nonlinear model has at least one parameter (quantity to be estimated) that appears nonlinearly [24,25]. Nonlinear regression is an iterative procedure, and the basis used for estimating the unknown parameters is the criterion of least squares. The fitting

procedure provides (i) parameters, (ii) error estimate on the parameter, and (iii) a statistical measure of goodness of fit.

The least squares criterion quantifies goodness of fit as the sum of squares of the vertical distances of the data points from the assumed model. That is, the best model for a particular data set is that with the smallest sum of squares. In fact, it is not simple to compare models with different parameters. The problem is that a more complicated model (more parameters) gives more flexibility (more inflection points) for the curve being generated than the curve being defined by a simpler model (fewer parameters). Thereby, the sum of squares of a more complicated model tends to be lower.

Both Collipriest and Priddle models use two parameters (C and n), while the modified Forman contains five parameters (C, n, m, p, and q). The objective of the exponent m is to control the spread of the FCG rate curve for different values of R. For the case of constant R, the exponent can be taken as zero, reducing the model parameters to four. In addition, p and q may be used as constants, reducing the model parameters to two. For metallic materials, since the values of p and q are between 0 and 1, they were chosen by trial end error to be 0.25 and 0.75, respectively. Finally, both C and n of the selected models may be estimated using nonlinear regression analysis.

3.3.2. Parameter estimation

The estimated parameters (C and n) and the statistical properties obtained for all pre-strain cases studied and models selected are presented in Tables 4–6. The estimated curves are presented in Figs. 4–6.

For the 0% pre-strain case, Fig. 4 shows that the modified Forman model provides a better fit to the FCG rate data as compared to the other two models. The Collipriest and Priddle models fit the data in a similar fashion; however, the latter fits relatively better than the former.

The estimated parameters (C and n) of the modified Forman model are the most significant parameters. This may be confirmed by examination of t-value and related p-value of

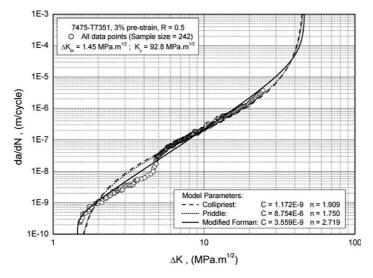


Fig. 5. FCG rate curves of 7475–T7351, 3% pre-strain, R=0.5, comparison among Collipriest, Priddle, and modified Forman models.

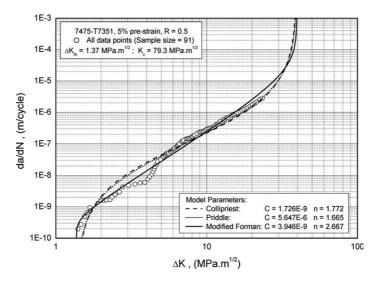


Fig. 6. FCG rate curves of 7475–T7351, 5% pre-strain, R=0.5, comparison among Collipriest, Priddle, and modified Forman models.

the estimates for each model (Table 4). The *t*-value is the ratio of the parameter estimate to its standard error, the latter being used to compute confidence intervals of the estimate. Generally, a high *t*-value associated with a parameter estimate indicates that the estimate is well determined in the model and the confidence band will be narrow. Conversely, a low *t*-value indicates that the estimate is poorly determined and the confidence band will be wide. An indication of high or low *t*-value of an estimate depends on the related *p*-value (probability) and significance level used. For a 0.05 significance level, an estimate is statistically significant when its *t*-value is high enough to provide a *p*-value less than the significance level chosen.

From Table 4, at the 0.05 and 0.01 levels of significance, it is concluded that the parameters (C and n) estimated from the three models are statistically significant, since their p-values are smaller than both levels of significance. The modified

Forman parameters are more significant due to their larger *t*-values.

3.3.3. Goodness of fit

There are several statistical measures that may be used to assist the goodness of fit of a model used with the experimental data. These are: R^2 , $R_{\rm adj}^2$, F-value, standard error of the residual (Se), Schwarz's Bayesian Information Criterion (BIC), and Akaike's Information Criterion (AIC) [36]. The R^2 statistic may be used for comparing models having the same number of parameters. Other statistical methods may be used for comparing models that have an equal or different numbers of parameters. Selection of the best model should be based on other considerations than just the value of one model selection statistic.

In this study, a comparison among the FCG rate models chosen was based on R^2 , $R_{\rm adj}^2$, F-value, and standard error of the residual (Se). Large values of R^2 , $R_{\rm adj}^2$, and F statistics coupled

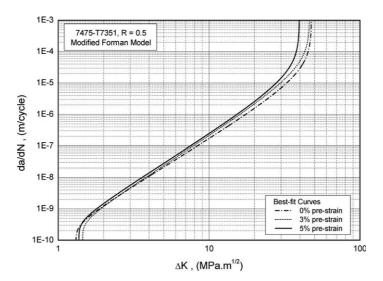


Fig. 7. Effect of pre-strain on the FCG rate of 7475-T7351, R=0.5, modified Forman model.

with a small value of Se indicate that the model explains the data well. From Table 4, all statistical measures determined from the modified Forman model indicate that this provides the best fit as compared with the other two models. Also, the Priddle model fits the data better than the Collipriest one.

A visual examination of the curves presented in Fig. 4 reveals that the modified Forman curve comes close to the data points in both the threshold and Paris regions as compared to the curves generated by Collipriest and Priddle models.

For the case of 3% pre-strain (Table 5 and Fig. 5) and 5% pre-strain (Table 6 and Fig. 6), the same results outlined for the case of 0% pre-strain were found. This implies that the modified Forman model fits the FCG rate data better than the Priddle model, which fits relatively better than the Collipriest model.

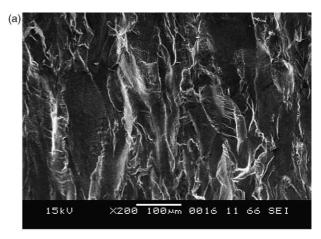
3.4. Effect of pre-strain on the FCG rate

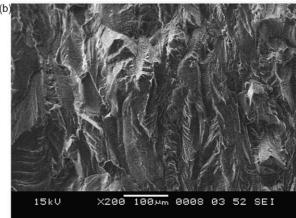
The modified Forman model provides a better fit to the FCG rate data when compared with the other two models. Hence, it is used to show the effect of pre-strain on FCG rate. Fig. 7 shows that an increase in pre-strain level has no appreciable influence in the near-threshold region (region I) and in the Paris region (region II). However, in the unstable region (region III), a considerable effect can be seen, where an increase in pre-strain from 0 to 5% leads to a decrease in material fracture toughness and consequently alters the FCG behavior.

The Paris region is less sensitive to microstructural alterations as a result of the pre-strain applied. On the other hand, if pre-strain can change the microstructure, the threshold value will be affected. There are several factors that influence the near-threshold FCG rate, like grain size [37,38], dispersoids [39], precipitates [40,41], texture [42]. However, in Ref. [43] no appreciable effect of the pre-straining on the microstructure of 7475–T7351 was found. Therefore, it may be expected that the micromechanisms of fatigue crack growth are similar for all cases of pre-strain.

The SEM examination of the fracture surfaces in the near-threshold region reveals that crack growth progressed by cyclic-cleavage with ductile striations (Fig. 8). Since there are no considerable differences in the severity of fatigue crack growth micromechanisms for the different pre-strain levels, it may be expected that the true mean threshold values are similar. This is shown in Table 3, where the differences in the threshold value for the three conditions did not exceed 10%.

The SEM examination of the fracture surfaces in the transition between threshold and Paris regions (Fig. 9) shows that the ductile striations are the dominant features. In this region, since there are no considerable differences in the severity of fatigue crack growth micromechanisms for the prestrain cases, it may be concluded, therefore, that the FCG rates are nearly the same.





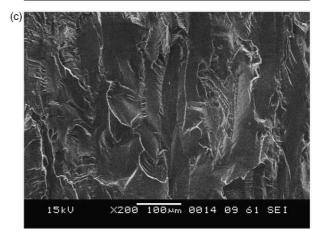
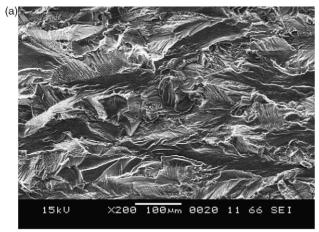


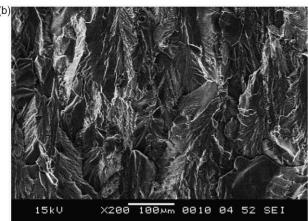
Fig. 8. SEM micrographs of the fracture surfaces in the near-threshold region, 7475–T7351. (a) 0% pre-strain, (b) 3% pre-strain, and (c) 5% pre-strain.

4. Conclusions

The FCG behavior of pre-strained 7475–T7351 aluminum alloy plate was studied. Two tensile pre-strain levels of 3 and 5% were applied. FCG tests were carried out at R = 0.5 and the data collected were modeled using Collipriest, Priddle, and modified Forman models. From the results obtained, the following conclusions can be drawn:

1. The effect of pre-strain may be clearly seen in the unstable region, region III, where the increase of pre-strain leads to





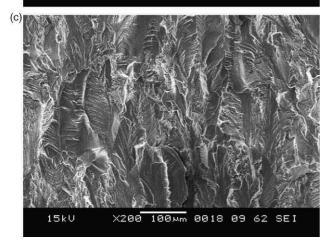


Fig. 9. SEM micrographs of the fracture surfaces in the transition between threshold and Paris regions, 7475–T7351. (a) 0% pre-strain, (b) 3% pre-strain, and (c) 5% pre-strain.

decrease material fracture toughness, thereby, increasing the FCG rates. The pre-strain has no considerable influence in the near-threshold region and in the Paris region.

- 2. Both Collipriest and Priddle models fit the FCG rate data sets in a similar fashion. However, the latter provides a better fit than the former one.
- The modified Forman model provides the best fit to the FCG rate data sets as compared with the Collipriest and Priddle models. Therefore, this model may be suggested for

use in critical applications, such as aeronautical structural design.

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